**MARTIAN GEOLOGY**

**Large wind ripples on Mars: A record of atmospheric evolution**

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Wind blowing over sand on Earth produces decimeter-wavelength ripples and hundred-meter–to kilometer-wavelength dunes: bedforms of two distinct size modes. Observations from the Mars Science Laboratory Curiosity rover and the Mars Reconnaissance Orbiter reveal that Mars hosts a third stable wind-driven bedform, with meter-scale wavelengths. These bedforms are spatially uniform in size and typically have asymmetric profiles with angle-of-repose lee slopes and sinuous crest lines, making them unlike terrestrial wind ripples. Rather, these structures resemble fluid-drag ripples, which on Earth include water-worked current ripples, but sloping active bedforms (Fig. 1B and fig. S3) (8). Grain-impact processes are thought to dominate the formation of wind ripples, whereas dune formation involves an aerodynamic instability (6). Orbital observations of Mars also show the superposition of two distinct scales of active bedforms (Fig. 1B and fig. S3) (8). Martian dunes form at a similar wavelength as on Earth; however, those dunes are ubiquitously mantled with bedforms 1 to 5 m in wavelength (hereafter referred to as large martian ripples) (9).

Large martian ripples were thought to have a similar origin to decimeter-wavelength eolian impact ripples on Earth, but to be larger on Mars because of differences in saltation (ballistic hopping of grains) (6). An implicit assumption under this hypothesis is that small wind ripples should not coexist with large martian ripples. Until recently, the spatial coexistence of three scales of bedforms could not be tested because the resolution of orbital imagery was too coarse (25 to 50 cm per pixel in

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Fig. 1. Eolian bedforms on Earth and Mars. (A) Dunes and ripples at Oceano Dunes, California, United States (35.094960°N, -120.623476°E). (B) to (F) Dune in the Bagnold dune field, Gale crater, Mars, as shown from (B) a HiRISE image (ESP_035917_1755) and [(C) to (F)] the Curiosity rover. (C) Mastcam mosaic (mcam05410, sol 1192) showing small and large ripples on the dune. (D) Mastcam image (mcam05600, sol 1221) of large ripples with superimposed small ripples. (E) MAHLI 25-cm standoff image (1223MH0005550010403094C00, sol 1223), ~1 m off frame of (D) in the direction of the black circle and arrow. (F) 5-cm standoff image (1223MH0005560010403097C00, sol 1223) of the crest of a large ripple.
High Resolution Imaging Experiment (HiRISE) images (10) to detect decimeter-scale ripples, and rovers had not visited active dune fields, only sand sheets and coarse-grained ripples (11, 12, 13). Observations made by the Curiosity rover (14) at an active dune field (the Bagnold dune field) (15) in Gale crater now show that large martian ripples are not simply larger versions of the decimeter-scale wind ripples seen on Earth. Rather, we observe decimeter-scale ripples superimposed on larger, meter-scale ripples, which are in turn superimposed on dunes (Fig. 1 and fig. S2). Thus, two stable ripple-scale bedforms coexist on Mars and both are superimposed on dunes, in contrast to the single scale of superimposed terrestrial ripples.

Curiosity images indicate that large martian ripples have morphologies unlike those of eolian impact ripples. Terrestrial impact ripples have sharp crests and asymmetric topographies with distinct upwind (stoss) and downwind (lee) slope angles. Furthermore, the stoss slopes of the large ripples are mantled by small-scale ripples with a wavelength range of ~5 to 12 cm, which, based on their sharp crests, we interpret as impact ripples similar to those of Earth (Fig. 1, C and D). This interpretation is consistent with recent numerical modeling that predicts that the large ripples should have decimeter-scale wavelengths (16). In contrast, the stoss crests of the large ripples are sharp and give way downslope to angle-of-repose slip faces (slopes dipping >30° downwind; fig. S5A) marked by the presence of grainflows (small avalanche deposits (Fig. 1D)), indicating recent activity. The presence of grainfall (i.e., sand that settles out on the lee slope) and deflected impact ripples on the lee slope indicates the aerodynamic influence of the large ripples contemporaneous with small-ripple migration (Fig. 1D).

We compiled a comprehensive multiscale data set of eolian bedform wavelengths on Mars by combining remote measurements from 11 martian sites (fig. S1 and tables S1 and S2), with rover measurements from stereo imagery in Gale crater (fig. S5) (7). Our statistical analysis confirms that Mars has an additional bedform-wavelength mode and that meter-scale ripples are absent in terrestrial eolian landscapes (Fig. 2 and table S3) (7). Large martian ripples are not simply small dunes, because they maintain a stable size, whereas meter-wavelength dunes, which are rare on Earth, grow as they translate downwind (6) (fig. S3 versus fig. S4). Large martian ripples mantled with impact ripples also cannot be explained as large versions of terrestrial impact ripples forming by large saltation (18, 19); no existing model can reproduce the coexistence and coevolution of two scales of impact ripples (20) (supplementary online text). Moreover, the large-ripple morphology differs significantly from that of impact ripples. An alternative interpretation of the large ripples is that they are coarse-grained ripples (21). However, images from the Mars Hand Lens Imager [MAHLI (16)] show well-sorted large ripples up the dune’s stoss slopes (Fig. 1E, with very fine to medium sand and no significant grain-size differences between the small and large ripples (Fig. 1, E and F). Thus, neither the impact nor the coarse-grained hypothesis readily explains the coexistence of two distinct equilibrium scales of active ripples composed of sediments of similar size.

Their stable size, sinuous crests, and asymmetric profiles with avalanche faces make the large martian ripples morphologically similar to terrestrial subaqueous current ripples (fig. S6), also called fluid-drag ripples (22) (supplementary text). If the large martian ripples form aerodynamically (i.e., wind-drag ripples (4, 23)), then they developed for current ripples should predict their scale once adjusted for martian conditions. Decades of flume experiments (24, 25) have led to scaling relations for current ripples (25, 26). Following the theoretical framework of (25), we cast ripple size data in terms of the dimensionless current ripple wavelength, \( \lambda^* = \frac{\lambda}{n} \) (where \( \lambda \) is ripple wavelength, \( n \) is kinematic fluid viscosity, \( u_e \) is bed shear velocity, and \( \lambda \) is proportional to the viscous sublayer thickness (25)), which is a function of the parameter \( R e_p \sqrt{\lambda} \) (where \( Re_p \) is particle Reynolds number and \( \sqrt{\lambda} \) is Shields stress (fig. S7 and supplementary text)). These dimensionless variables provide a complete description of ripple-size scaling that accounts for fluid and grain properties and for gravity. A large database of current ripple wavelengths (25), updated here to include results from high-viscosity fluids (24), illustrates that

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\lambda^* = 2453 (Re_p \sqrt{\lambda})^{1/4} \tag{1}
\]

(Fig. 3). To compare the predictions of fluid-drag ripple wavelengths to the large martian ripples, we calculated \( \lambda^* \) and \( \lambda^* \) for all compiled martian bedforms (7) (supplementary text). Results show that wind-drag ripples on Mars are predicted to be much larger than the decimeter-scale impact ripples because of the high kinematic viscosity in Mars’ low-density atmosphere; furthermore, the wavelength of the large martian ripples is consistent with fluid-drag theory (Fig. 3) across a range of elevations with different atmospheric density (fig. S10).

Because wind-drag ripples are predicted to be smaller in thicker atmospheres, identification of these bedforms in ancient sedimentary rocks (27) offers the potential to reconstruct atmospheric loss and the global drying of Mars (28). The migration of bedforms produces cross-stratification in sedimentary rocks, which can be used to determine their original three-dimensional geometry. Based on morphology and scale, and using a kinematic model (29) (Fig. 4), we expect sinuous wind-drag ripples formed under present-day martian atmospheric conditions (30) to form decimeter-thick trough cross-sets, grouped into larger sets formed by overall migration of the dune (supplementary text). Large-ripple stratification should be distinct from that of compound wind dunes or coarse-grained ripples, because compound dunes do not maintain a persistently stable size in the down-dip direction (fig. S3 versus fig. S4) and typically form thicker cross-sets, and coarse-grained ripples leave recognizable coarse-grained lags. Stratification from the large ripples might best resemble that of subaqueous ripples and dunes. However, identification of distinctive wind-ripple strata (inversely graded millimeter-thick continuous layers (31)) coexisting with both decimeter-scale cross-sets and meter-scale dune troughs would enable the definitive interpretation of an eolian origin, whereas other contextual support, such as fluvial bar sets, desiccation cracks, and soft-sediment deformation, would characterize wet environments (27).

Candidate wind-drag ripples were observed by the Opportunity rover at Cape St. Mary, Victoria crater, in the Burns formation (Fig. 4 and supplementary text) (32) and were recognized as
In contrast to martian dunes (pink squares) and small martian ripples (orange triangles), large martian ripples [red diamonds, n = 7280 bedforms, measured over 36 locations globally, including our measurements (7) and those of (38); the red star indicates rover measurements at Gale crater] match fluid-drag ripple theory. Symbols are means and error bars represent standard deviations at a given measurement site; error bars are smaller than marker size where not shown.

**REFERENCES AND NOTES**


7. Materials and methods are available as supplementary materials on Science Online.


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A vacuum flash-assisted solution process for high-efficiency large-area perovskite solar cells

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Metal halide perovskite solar cells (PSCs) currently attract enormous research interest because of their high solar-to-electric power conversion efficiency (PCE) and low fabrication costs, but their practical development is hampered by difficulties in achieving high performance with large-size devices. We devised a simple vacuum flash-assisted solution processing method to obtain shiny, smooth, crystalline perovskite films of high electronic quality over large areas. This enabled us to fabricate solar cells with an aperture area exceeding 1 cm², a maximum efficiency of 20.5%, and a certified PCE of 19.6%. By contrast, the best certified PCE to date is 15.6% for PSCs of similar size. We demonstrate that the reproducibility of the method is excellent and that the cells show virtually no hysteresis. Our approach enables the realization of highly efficient large-area PSCs for practical deployment.

In the span of a few years, the power conversion efficiency (PCE) of perovskite solar cells (PSCs) has risen from 3.8% (1) to 22.10% (2), which is unprecedented in the field of photovoltaics. However, such high efficiencies have been achieved only with cells of very small size—between 0.04 and 0.2 cm²—and few investigators have attempted to fabricate larger-area cells (3–10). The use of small-area devices has raised some doubts on the remarkable progress of the PSC field because the measurement errors tend to increase as the active cell area becomes smaller. Thus, the development of PSCs with a mandatory minimum active area of >1 cm² is required for the evaluation of this new photovoltaic (PV) technology (3, 8). At present, the best certified PCE of a cell with a size exceeding the critical threshold of 1 cm² is 15.6% (11) because of the limitations of current preparation methods (12–15). Performing PSCs often are made with an antisolvent such as chlorobenzene to precipitate the perovskite or its intermediate from its solution, which typically contains a solvent mixture of γ-butyltoluene (GBL), dimethylformamide (DMF), and dimethylsulfoxide (DMSO) (3, 7, 8, 14, 16–19). The antisolvent induces oversaturation of the perovskite solution, but because the antisolvent is usually dripped in the center of the film during spin-coating, the result is a radial gradient in oversaturation; the spatially inhomogeneous nucleation of the perovskite or its intermediate ultimately leads to defects in the perovskite film (3, 20). In addition, the antisolvents currently used are toxic and harmful to the environment, hampering their large-scale application (8, 14). Thus, alternative procedures for preparing large-area PSCs are warranted if their performance on a large device area is to be competitive with that of inorganic thin-film photovoltaics.

We developed a simple and effective method to produce high-quality perovskite films for large-area PSCs by applying a vacuum-flash treatment during the solution processing of the perovskite. Our approach differs from previous studies that used high-vacuum methods for vapor deposition of the perovskite (33, 35) or for removal of reaction products (i.e., methylammonium chloride) by sublimation during the thermal annealing of the films (27). Vacuum flash-assisted solution processing (VASP) is a method that enables the sudden and well-controlled removal of solvent, thereby boosting rapid crystallization of a fibrous material that consists of a Lewis acid-base-type adduct representing the perovskite precursor phase (19). Upon thermal annealing, the precursor phase produces highly oriented, crystalline perovskite films of excellent electronic quality that can be grown on a variety of substrates (17–19). Furthermore, VASP allows deposition of the perovskite films on large substrate sizes and can be turned into a continuous process.

We achieved a maximum PCE of 20.5% and a certified PCE of 19.6% for large cells with square aperture areas greater than 1 cm², which is commensurate with the 21.0% reached by today’s best thin-film copper indium gallium selenide (CIGS) and CdTe devices of similar size (12). Our method also eliminates the hysteresis in the current–voltage (J–V) curves, a notorious problem with PSCs (22). We tested the method with state-of-the-art perovskites using formamidinium (FA) and methylammonium (MA) mixed-cation and iodide-bromide mixed-anion perovskite formulations (16, 23) of composition FA0.81MA0.15PbI2.51Br0.45–0.55. To demonstrate the versatility of the method, we also prepared the emerging cesium (Cs+) and FA mixed-cation perovskite FA-Cs0.8PbI2.5Br–0.5 films (24–26). The VASP method is also readily scalable to the industrial level.

The basic steps of perovskite film fabrication by the VASP method are shown in Fig. 1A. The perovskite precursor solution, of composition FA0.81MA0.15PbI2.51Br0.45, containing DMSO with a nominal 1:1 ratio of lead to DMSO, was first spin-coated on top of a mesoporous TiO2 film prepared as described (16). The film was then placed for a few seconds into a vacuum chamber to boost rapid crystallization of the perovskite intermediate phase by removing most of the residual solvents, consisting mainly of GBL and DMF. We observed that the pressure applied...
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